

# New high-yield preparation procedure of $\text{Ln}[\text{Fe}(\text{CN})_6] \cdot n\text{H}_2\text{O}$ ( $\text{Ln} = \text{La}, \text{Gd}, \text{and Lu}$ ) and their thermal decomposition into perovskite-type oxides

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## Abstract

The heteronuclear  $\text{Ln}[\text{Fe}(\text{CN})_6] \cdot n\text{H}_2\text{O}$  ( $\text{Ln} = \text{La}, \text{Gd}, \text{and Lu}$ ) complexes were synthesized by the mixing of  $(\text{NH}_4)_3[\text{Fe}(\text{CN})_6]$  and  $\text{Ln}(\text{NO}_3)_3$  methanol solutions as a new method. Their thermal decomposition products were then investigated. Although the yield of the  $\text{Ln}[\text{Fe}(\text{CN})_6] \cdot n\text{H}_2\text{O}$  complexes was ca. 20% for the well-known method in a water solution, it was improved to ca. 90% using this new method. The X-ray diffraction (XRD) patterns of the complexes agreed with those of the  $\text{Ln}[\text{Fe}(\text{CN})_6] \cdot n\text{H}_2\text{O}$  single phase for all the examined Ln systems. The  $\text{LnFeO}_3$  single phase was obtained by the thermal decomposition of the synthesized heteronuclear complexes for  $\text{Ln} = \text{La}, \text{Gd}, \text{and Lu}$ .

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## 1. Introduction

Perovskite-type oxides are promising materials for many kinds of applications, such as chemical sensors, electrodes and electrolytes for fuel cells, and catalysts [1–3]. In general, polymetallic oxides have been prepared by the conventional solid reaction method. Chemical processing methods, such as the sol–gel method are used to obtain fine powders. As a new method, we proposed the thermal decomposition of heteronuclear complexes for the preparation of the di- or tri-metallic oxides. The heterometallic oxides with relatively high specific surface areas were formed at low temperatures, when the heteronuclear complexes were used as the precursors [4–10]. The decomposition of the heteronuclear  $\text{Ln}[\text{Fe}(\text{CN})_6] \cdot n\text{H}_2\text{O}$  ( $\text{Ln} = \text{lanthanoid}$ ) hexacyano-complexes

is a promising method for the preparation of homogeneous mixed oxides on an atomic level [11]. Up to now, the heteronuclear hexacyano-complexes have been prepared using the aqueous solutions mixture of  $\text{K}_3[\text{Fe}(\text{CN})_6]$  and the lanthanoid salt [11–14]. However, we have not obtained the heteronuclear complexes in high yield. We postulated that the yield of the complexes should improve, when a methanol solution is used instead of the water solution, since the solubility of the complex in methanol is lower than that in water. However, the usual  $\text{K}_3[\text{Fe}(\text{CN})_6]$  could not be used as a starting material because of its low solubility in methanol. From this point of view, we considered that the utilization of  $(\text{NH}_4)_3[\text{Fe}(\text{CN})_6]$  having a high solubility in methanol as the starting material would produce high yields of the  $\text{Ln}[\text{Fe}(\text{CN})_6] \cdot n\text{H}_2\text{O}$  complexes.

In this study, we prepared the  $\text{Ln}[\text{Fe}(\text{CN})_6] \cdot n\text{H}_2\text{O}$  ( $\text{Ln} = \text{La}, \text{Gd}, \text{and Lu}$ ) complexes using  $(\text{NH}_4)_3[\text{Fe}(\text{CN})_6]$  with methanol. Their decomposition products were then stud-

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ied by thermogravimetry (TG) analysis and X-ray diffraction (XRD) to confirm the formation of the Ln:Fe = 1:1 complexes.

## 2. Experimental

We compared two sample preparation processes, i.e., for previous one (Method 1) and a new method (Method 2). For Method 1, the heteronuclear complex,  $\text{Ln}[\text{Fe}(\text{CN})_6] \cdot n\text{H}_2\text{O}$ , was synthesized by mixing aqueous solutions of equimolar amounts of  $\text{K}_3[\text{Fe}(\text{CN})_6]$  ( $1.0 \text{ mol dm}^{-3}$ ) and  $\text{Ln}(\text{NO}_3)_3 \cdot n\text{H}_2\text{O}$  ( $1.0 \text{ mol dm}^{-3}$ ) under continuous stirring at room temperature. The resulting precipitate was collected by suction filtration and then washed with water, ethanol, and diethyl ether, before drying in air at  $50^\circ\text{C}$ . For Method 2, the heteronuclear complex was synthesized by mixing 4:1 methanol solutions of equimolar amounts of  $(\text{NH}_4)_3[\text{Fe}(\text{CN})_6]$  ( $0.25 \text{ mol dm}^{-3}$ ) and  $\text{Ln}(\text{NO}_3)_3 \cdot n\text{H}_2\text{O}$  ( $1.0 \text{ mol dm}^{-3}$ ) under continuous stirring at room temperature for few min. The resulting precipitate was collected by suction filtration and then washed with ethanol and diethyl ether before drying in air at  $50^\circ\text{C}$ . The  $(\text{NH}_4)_3[\text{Fe}(\text{CN})_6]$  as the starting material of Method 2 can be obtained by different routes: (1) the mixed solution of the  $\text{K}_3[\text{Fe}(\text{CN})_6]$  methanol suspension and  $\text{NH}_4\text{Cl}$  methanol solution (1:3 mol ratio) was separated into the KCl precipitate and the filtered solution; the filtered solution was concentrated to obtain the  $(\text{NH}_4)_3[\text{Fe}(\text{CN})_6]$  material and (2) the water suspension of  $\text{K}_3[\text{Fe}(\text{CN})_6]$  ( $1.0 \text{ mol dm}^{-3}$ ) and  $(\text{NH}_4)_2\text{C}_2\text{O}_4$  powder (1:1.5 mol ratio) were mixed for 2 days, then separated into the  $\text{K}_2\text{C}_2\text{O}_4$  precipitate and the filtered solution. The  $(\text{NH}_4)_3[\text{Fe}(\text{CN})_6]$  was obtained from the concentrated solution. In our case, we used method (2). In addition, we confirmed that  $(\text{NH}_4)_3[\text{Fe}(\text{CN})_6]$  was also obtained using the  $\text{La}[\text{Fe}(\text{CN})_6] \cdot 5\text{H}_2\text{O}$  complex ( $1.0 \text{ mol dm}^{-3}$ ) instead of  $\text{K}_3[\text{Fe}(\text{CN})_6]$  to form the  $\text{La}_2(\text{C}_2\text{O}_4)_3$  precipitate.

The complex was decomposed at selected temperatures (in the range from  $400$  to  $1000^\circ\text{C}$ ) for 1 h, in order to obtain the perovskite-type powders. In order to characterize the complexes and the decomposition products, their X-ray diffraction patterns using  $\text{Cu K}\alpha$  radiation were recorded (Rint 2000, Rigaku, scanning rate =  $2^\circ/\text{min}$  at  $40 \text{ kV}$  and  $20 \text{ mA}$ ). The thermal decomposition behavior was examined by TG analysis, performed at a heating rate of  $5^\circ\text{C}/\text{min}$  in air.

## 3. Results and discussion

The  $\text{Ln}[\text{Fe}(\text{CN})_6] \cdot n\text{H}_2\text{O}$  complexes have been synthesized using our previous method (Method 1). The yield of the complexes was ca. 20% for all the Ln systems. With this method, it is very difficult to improve the yield using a highly concentrated solution, because the concentration of the  $\text{K}_3[\text{Fe}(\text{CN})_6]$  aqueous solution ( $1.0 \text{ mol dm}^{-3}$ ) as the starting material is almost at the maximum solubility limit.

Table 1  
Yield of  $\text{Ln}[\text{Fe}(\text{CN})_6] \cdot n\text{H}_2\text{O}$  (Ln = La, Gd, and Lu) complexes using Method 2

| Ln | <i>n</i> | Yield (%) |
|----|----------|-----------|
| La | 6        | 92        |
| Gd | 4        | 94        |
| Lu | 4        | 88        |

Hydrated *n* value was estimated using TG measurement.

Table 1 presents the yield of the complexes using the new method (Method 2). The yield was improved to ca. 90% for all the examined complexes. This high yield also suggests similar results for the other kinds of Ln-complexes. In this case, the *n* value in the complexes was estimated by the TG measurement. Fig. 1 shows the TG/DTA result of the  $\text{La}[\text{Fe}(\text{CN})_6] \cdot n\text{H}_2\text{O}$  complex. Dehydration by the loss of water of crystallization started at about  $80^\circ\text{C}$ . A weight loss due to the exothermal decomposition of the ligand (confirmed by DTA) was confirmed at around  $320^\circ\text{C}$  followed by a slower loss that ended at about  $720^\circ\text{C}$ . The weight loss percentage of 51.5 wt% in the last plateau range gave ca. *n* = 6, which was greater than 55.0% for the general formula of the La-complex with five molecules of water tied up in the crystallization [12]. The complex and decomposed products were investigated by XRD are shown in Fig. 2. The XRD peak position for the complex agreed with that of  $\text{La}[\text{Fe}(\text{CN})_6] \cdot 5\text{H}_2\text{O}$ , which has a hexagonal structure (space group:  $P6_3/m$ ) [12]. In addition, the decomposed product at  $400$  and  $500^\circ\text{C}$  showed the peaks of the  $\text{LaFeO}_3$  phase with a “halo” at around  $30^\circ$  without those of the mono-metallic oxide phase, such as  $\text{La}_2\text{O}_3$  and  $\text{Fe}_2\text{O}_3$ . The single perovskite phase was obtained at  $600^\circ\text{C}$  or higher. This formation of the  $\text{LaFeO}_3$  phase means that the synthesized material was pure  $\text{La}[\text{Fe}(\text{CN})_6] \cdot 5\text{H}_2\text{O}$  even for this high-yield method. The water absorption from atmosphere might be a cause for the

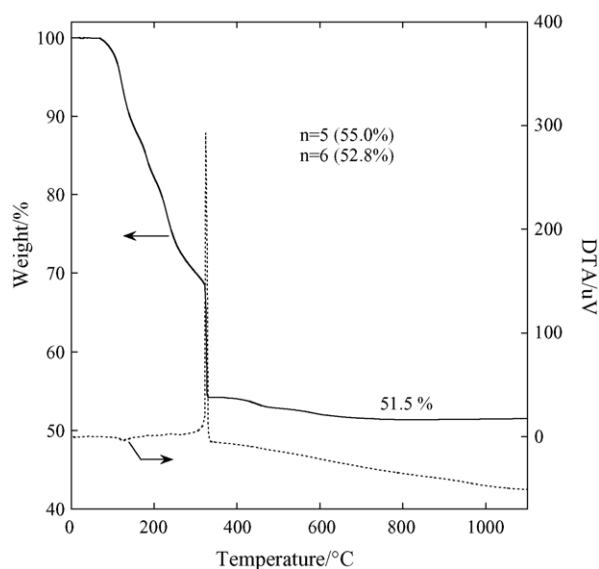


Fig. 1. TG/DTA result for  $\text{La}[\text{Fe}(\text{CN})_6] \cdot n\text{H}_2\text{O}$  complex (Method 2). Weights of experimental and (theoretical) of *n* = 5 and 6 are shown.

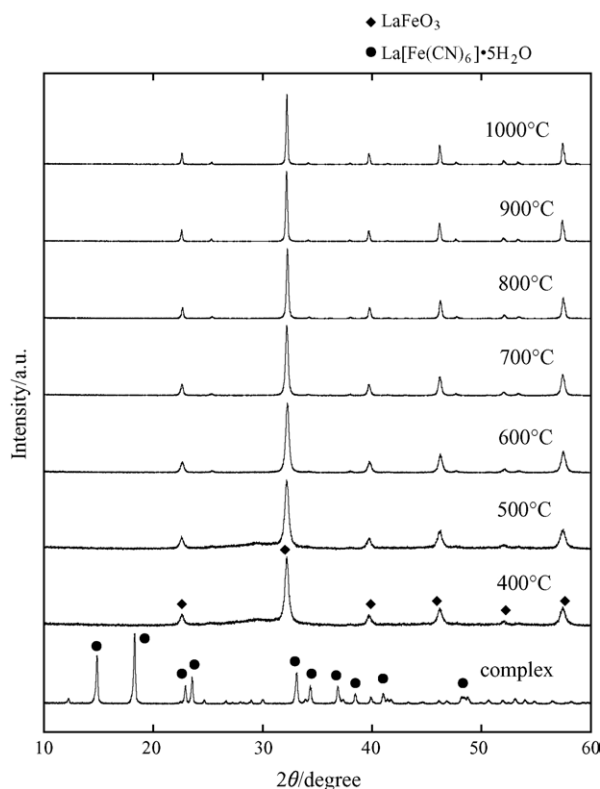


Fig. 2. XRD results of La[Fe(CN)<sub>6</sub>] $\cdot$ *n*H<sub>2</sub>O complex (Method 2) and decomposed products. Decomposed temperature is shown.

higher *n* value in the TG result for the LaFe-complex. It is confirmed that the full-width-at-half-maximum (FWHM) of all the LaFeO<sub>3</sub> peaks decreased with an increase in the decomposition temperature up to 1000 °C because of the crystallite growth.

For Ln = Gd and Lu, 58.5 and 60.1% for TG measurements of the decomposed materials almost agreed with the theoretical values of 59.1 and 60.6% as *n* = 4, respectively. Figs. 3 and 4 show the XRD results for the complex and decomposed products of Ln = Gd and Lu, respectively. The XRD peaks for each of the complexes showed an orthorhombic structure (space group: *CmCm*), which clearly agreed with the reported crystal having *n* = 4 [12]. For the decomposed products of Ln = Gd, the XRD patterns of the samples fired at 400–600 °C showed “halo” and no peaks. The peaks of the perovskite-type phase were detected from the samples decomposed at 700 °C. The GdFeO<sub>3</sub> single phase was obtained from the samples decomposed at or above 800 °C. In our previous paper, we reported that the single perovskite phase was formed by the thermal decomposition of the complexes for Ln = La, Pr, Nd, Sm, and Gd in the 600–900 °C in temperature range [4]. The temperature for the formation of the perovskite-type oxide gradually increased with an increase in the atomic number for Ln, i.e., a decrease in the ionic radius of the Ln<sup>3+</sup> ion for Ln = La, Pr, Nd, Sm, and Gd. For the Ln = Gd system, the formation temperature (800 °C) of the single perovskite-type phase clearly agreed

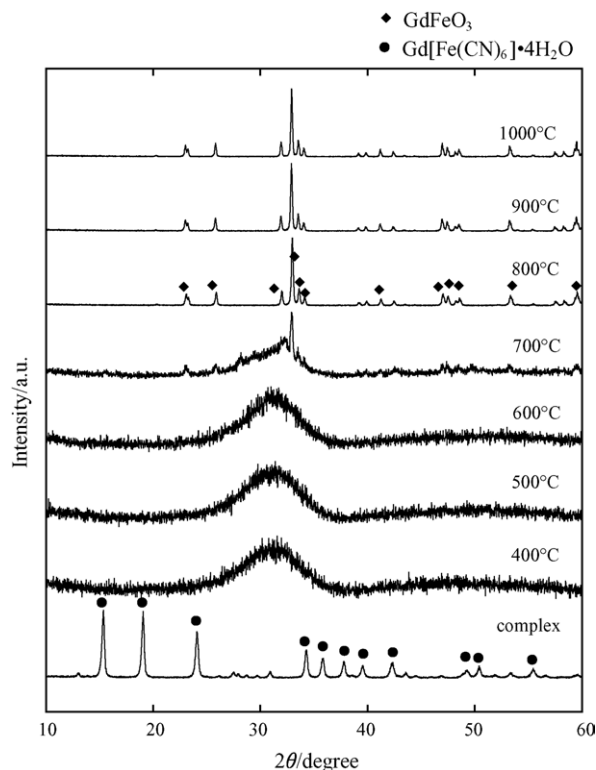


Fig. 3. XRD results of Gd[Fe(CN)<sub>6</sub>] $\cdot$ *n*H<sub>2</sub>O complex (Method 2) and decomposed products. Decomposed temperature is shown.

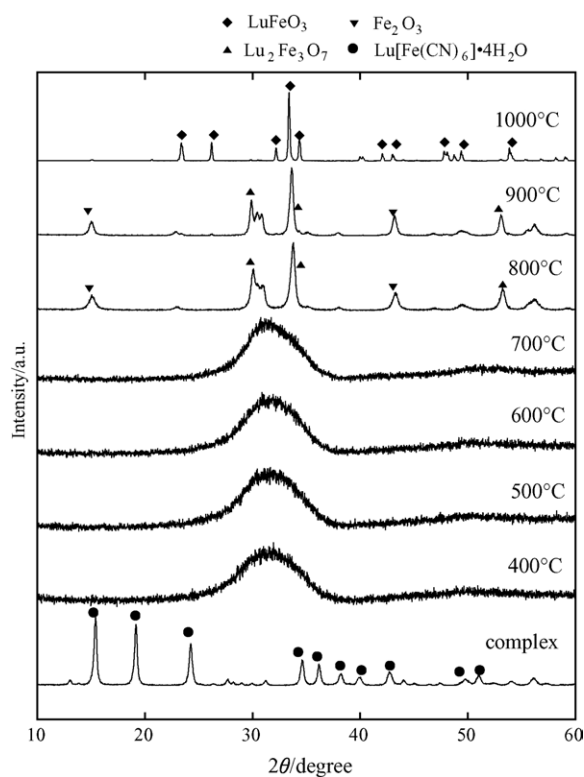


Fig. 4. XRD results of Lu[Fe(CN)<sub>6</sub>] $\cdot$ *n*H<sub>2</sub>O complex (Method 2) and decomposed products. Decomposed temperature is shown.

with that of our previous results [4]. The single phase of the perovskite-type oxide did not form for the heavy Ln=Dy and Ho of the decomposed samples even at 900 °C [4]. For Ln=Dy, the DyFeO<sub>3</sub> single phase was obtained by sintering at 1000 °C [1]. Up to now, the thermal decomposition products have not been investigated for the heavy Ln=Er, Tm, Yb, and Lu systems. Although the Lu<sup>3+</sup> ion has the smallest ionic radius in the lanthanoid, the formation of the single perovskite LuFeO<sub>3</sub> phase was confirmed for the decomposed sample even at 1000 °C. From the XRD results for the complex and the decomposed material, it is very clear that the Ln:Fe ratio was 1:1 for all the samples synthesized by the new high-yield method (Method 2) using methanol.

#### 4. Conclusions

The perovskite-type oxides are some of the most useful materials for many applications. We have been investigating the preparation of perovskite-type oxide powders having a high homogeneity and small particle size by the thermal decomposition of the appropriate hexacyano Ln[M(CN)<sub>6</sub>] $\cdot$ nH<sub>2</sub>O heteronuclear complexes. In this study, a new method was discovered for the high-yield (ca. 90%) preparation of the Ln[Fe(CN)<sub>6</sub>] $\cdot$ nH<sub>2</sub>O complexes by the mixing of (NH<sub>4</sub>)<sub>3</sub>[Fe(CN)<sub>6</sub>] and Ln(NO<sub>3</sub>)<sub>3</sub> in methanol. This synthesis method is very useful for applications,

such as chemical sensors and electrode materials for fuel cells.

#### References

- [1] H. Aono, E. Traversa, M. Sakamoto, Y. Sadaoka, *Sens. Actuators B* 94 (2003) 132–139.
- [2] N.Q. Minh, *J. Am. Ceram. Soc.* 76 (1993) 563–588.
- [3] J.G. McCarty, H. Wise, *Catal. Today* 8 (1990) 231–248.
- [4] Y. Sadaoka, K. Watanabe, Y. Sakai, M. Sakamoto, *J. Alloys Compd.* 224 (1995) 194–198.
- [5] E. Traversa, M. Sakamoto, Y. Sadaoka, *J. Am. Ceram. Soc.* 76 (1996) 1401–1404.
- [6] Y. Sadaoka, E. Traversa, M. Sakamoto, *J. Alloys Compd.* 240 (1996) 51–59.
- [7] Y. Sadaoka, E. Traversa, M. Sakamoto, *J. Mater. Chem.* 6 (1996) 1355–1360.
- [8] M. Sakamoto, P. Nunziante, E. Traversa, S. Matsushima, M. Miwa, H. Aono, Y. Sadaoka, *J. Ceram. Soc. Jpn.* 105 (1997) 963–969.
- [9] Y. Sadaoka, H. Aono, E. Traversa, M. Sakamoto, *J. Alloys Compd.* 278 (1998) 135–141.
- [10] H. Aono, S. Nakano, N. Kondo, H. Katagishi, M. Sakamoto, E. Asato, Y. Sadaoka, *Chem. Lett.* 2002 (6) (2002) 568–569.
- [11] H. Aono, K. Kinoshita, Y. Sadaoka, M. Sakamoto, *J. Ceram. Soc. Jpn.* 106 (1998) 958–963.
- [12] F. Hulliger, M. Landolt, H. Vetsch, *J. Solid State Chem.* 18 (1976) 283–291.
- [13] D.F. Mullica, W.O. Milligan, T. Kouba, *J. Inorg. Nucl. Chem.* 41 (1979) 967–973.
- [14] X. Wang, Y. Yukawa, Y. Masuda, *J. Alloys Compd.* 316 (2001) 107–112.